-0 D-The focus of this study is on the Fe_2O_3 -doped Bi_{0.5}Na_{0.5}TiO₃-SrTiO₃ piezoelectric material. This is important to find ecologically acceptable piezoelectric materials. This research aims to obtain a leadfree piezoelectric material because lead is a material that is not environmentally friendly. An alternative solution is a piezoelectric material based on BNT-ST, which in this case is doped with Fe_2O_3 material. The study of Fe₂O₃ doped Bi_{0.5}Na_{0.5}TiO₃-SrTiO₃ piezoelectric material prepared by the solid-state reaction method was carried out to determine the optimum composition of the material formed. Doping variations are 0; 2.5; 5; 7.5; and 10 in mol %. The examinations were performed using X-ray diffraction (XRD) spectroscopy, a Scanning Electron Microscope (SEM), and an LCR meter. The Fe₂O₃ doped Bi_{0.5}Na_{0.5}TiO₃-SrTiO₃produced a new compound in the form of FeBi₅Ti₃O₁₅-Na₂Ti₃O₇-SrTiO₃ with the crystal structure of cubic, orthorhombic, and monoclinic, as well as the increasing crystalline size with the addition of dopants, exclude at 5 mol % and 7.5 mol %. FeBi₅Ti₃O₁₅-Na₂Ti₃O₇-SrTiO₃ also produces varying particle sizes, which are between 0.88-8.23 µm. From the obtained data, the optimum composition of Fe_2O_3 doped Bi_{0.5}Na_{0.5}TiO₃-SrTiO₃ was the 2.5 mol % of Fe_2O_3 due to it having the highest dielectric constant (ε_r) and temperature Curie (T_c) , and also the lowest material impedance (Z) with the ε_r of 12.037 at T_c of 400 °C and Z of 135 k Ω . The high piezoelectricity, which is indicated by the high value of the dielectric constant and Curie temperature, is possible due to the presence of a greater number of sodium ions in the Na₂Ti₃O₇ phase. Sodium ions are ions with good electrical storage capabilities. The increase in dielectric constant in the BNT-ST piezoelectric obtained by the addition of Fe_2O_3 shows that this material can be used as a substitute for lead-based piezoelectric materials so that it is secure for the environment. The piezoelectric material of BNT-ST doped with Fe₂O₃ earned from this research can be applied to obtain electricity with a optimal value when given mechanical pressure

Keywords: piezoelectric, BNT-ST, Fe₂O₃ doping, electric properties, Curie temperature UDC 621

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IMPROVEMENT THE DIELECTRIC AND IMPEDANCE PROPERTIES OF PB-FREE Bi_{0.5}Na_{0.5}TiO₃-SrTiO₃ PIEZOELECTRIC MATERIALS MODIFIED BY Fe₂O₃

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1. Introduction

Piezoelectric technology is one of the new Renewable Energy technologies that has been widely used as a source of electrical energy. The working principle of piezoelectricity is a reversible reaction that produces an electric current due to pressure/strain. Today's most widely used piezoelectric material is Lead Zirconate Titanate (Pb(Zr_xTi_{1-x})O₃) commonly referred to as PZT [1–5]. One of the critical drawbacks of PZT is that it contains more than 60 % lead (Pb) by weight. This effective lead content creates a hazard during processing (lead evaporates and is released into the atmosphere) and limits applications (e. g. in vitro). It is potentially toxic to the environment during disposal. Over the last few years, the World Health Organization (WHO) has begun to impose strict restrictions on the use of lead, except for the electronics industry, due to the lack of PZT-compliant replacements [6]. Therefore, the researchers began to look for PZT replacement materials with more environmentally friendly materials.

One of the environmentally friendly piezoelectric materials that have recently become a concern for researchers is $Bi_{0.5}Na_{0.5}TiO_3$. Studies on the utilization and modification of $Bi_{0.5}Na_{0.5}TiO_3$ materials to produce increased piezoelectricity have been carried out, including modifying them with $Bi_{0.5}K_{0.5}TiO_3$, $SrTiO_3$ [5, 7–12]; $K_{0.5}Na_{0.5}NbO_3$ -Gd₂O₃ [13]; and $BaTiO_3$ -NaNbO₃ [14]. $Bi_{0.5}Na_{0.5}TiO_3$ has good piezoelectric properties with a Curie temperature of 320 °C, but this value is still lower than the Curie temperature of PZT. The Curie temperature value of a material affects the quality of the ma-

terial. Next, the piezoelectricity of the material is also influenced by the dielectric properties of the material, whose value is represented by the dielectric constant. Dielectric materials are vital in ferroelectricity because of some of the properties of dielectric materials; namely, they can store electric charge, pass alternating current, and withstand direct current.

Increasing the Curie temperature and dielectric constant can be done by substituting the $Bi_{0.5}Na_{0.5}TiO_3$ crystal lattice with ferrite (Fe₂O₃), which raises the formed structure's crystallinity, increasing the ceramic grain size, and increases the magnetoelectric response [15]. Modification of piezoelectric materials has been reported to be successful in increasing the Curie temperature and dielectric constant. Piezoelectric $Bi_{0.5}Na_{0.5}TiO_3$ (BNT) composited using SrTiO₃ (ST) and modified using Fe₂O₃ is more environmentally friendly, stable, and has the potential to produce better piezoelectricity values. Therefore, research devoted to the development of a lead-free piezoelectric material based on BNT-ST and modified by Fe₂O₃ is relevant.

2. Literature review and problem statement

Piezoelectric materials are a class of materials that can be polarized using mechanical pressure or an electric field. Examples of piezoelectric materials are Bismuth Natrium Titanate (BNT), Bismuth Natrium Titanate-Strontium Titanate (BNT-ST), Barium Titanate (BaTiO₃), Lead Titanate (PbTiO₃), Lead Zirconate Titanate (PZT), Lead Lanthanum Zirconate Titanate (PLZT), Lead Magnesium Niobate (PMN), Potassium Niobate (KNbO₃), etc. Three parameters can indicate the piezoelectric properties of a material, namely dielectric constant, Curie temperature, and material impedance [16].

Piezoelectric Bi_{0.5}Na_{0.5}TiO₃-SrTiO₃ (BNT-ST) is an environmentally friendly electrical energy-producing material compared to the PbZr_(0.52)Ti_(0.48)O₃ (PZT) material which is currently frequently used. PZT has a piezoelectric constant value d_{33} =460 PC/N, coupling factor K_p =0.56, P_t =39.2 µC/cm², coercive field E_c =14.9 kV/cm, and a high Curie temperature of 450 °C [17]. However, PZT contains the element Pb (lead) which is toxic. Industrial scale production requires quite a lot of Pb material so it is not profitable because waste gases such as lead oxide (PbO) evaporate and likewise the remaining Pb oxide material in the synthesis process can pollute the environment and be dangerous for health.

To overcome this concern, research and development of lead-free piezoelectric materials has been carried out, one of which is Bi_{0.5}Na_{0.5}TiO₃. (BNT). It is necessary to try to modify this material to obtain a material with the desired characteristics. The material modification process can be done by modifying the manufacturing process as has been done by several researchers [18, 19]. In this case, modification is carried out using microwave irradiation [18] and variations in milling time duration [19]. Or by providing additional ingredients known as a doping process, as has been done by other researchers [20–22]. The paper [20] presents the results of research related to the addition of Nd to Strontium hexaferrite. It is demonstrated that the structure, magnetic characteristics, and microwave absorption capability of strontium hexa ferrite change when Nd is added. On the other hand, research on the addition of Ba to BNT is presented in publication [21]. The structure and dielectric characteristics of the material generated are altered by the addition of Ba to BNT. Additionally, research on the topic of adding carbon black to a mixture of polybutylene terephthalate, polyamide 6, and carbon black is reported in the paper [22]. Here, the material's microstructure and mechanical characteristics are altered by the addition of carbon black. The results of the investigation showed that the addition of a material can change its structure, mechanical properties, and dielectric properties. However, questions related to the modification of BNT material, or the effects of adding doping material to BNT have not been revealed in detail. This may be because of the financial constraints and the fact that this topic is beyond the scope of their research focus.

In the development of BNT materials, many doping materials have been added in fairly good quantities in the hope of forming solid solutions that have mechanical and electrical properties comparable to PZT. Research on modifying BNT by adding ST doping (SrTiO₃) was carried out by Werner Krauss using the solid-state reaction method, obtaining a Curie temperature value of $335 \,^{\circ}$ C [23]. In making these materials, polyvinyl alcohol can be added to improve the mechanical properties of the material so that it has strong resistance when subjected to high pressure or temperature [24]. The higher the Curie temperature, the better the quality of the piezoelectric material.

Of several ferroelectric doping materials, one of them is Fe_2O_3 . Fe_2O_3 material is an ion donor doping material [25]. The use of this doping material is because Fe_2O_3 has a doping ion whose radius is almost the same as the Ti^{4+} ion. Fe_2O_3 has a Fe atom with a 3+ ion whose basic salt color is yellowish brown. The radius of the Fe^{3+} doping ion is 0.67, while the radius of the Ti^{4+} doping ion is 0.68 which occupies position B in the perovskite structure. The addition of the Fe^{3+} doping ion will result in the formation of space at position B in the ABO₃ structure.

The Fe₂O₃ addition in the solid-state reaction can increase the density and increase the piezoelectric constant [26]. They also reported that FeO and Fe₂O₃ doping could increase the grain size of KNN-based ceramics. In addition, many researchers have systematically studied the structural, magnetic, electrical, and magnetoelectric behavior of Fe³⁺ ions doped in perovskite ferroelectric ceramics (BaTiO₃, SrTiO₃, and KTaO₃) and these compounds show possible magnetoelectric response at room temperature [27]. So, Fe₂O₃ doping in appropriate proportions is an effective method to improve the densification and electrical properties and produce ferromagnetism in perovskite ferroelectric ceramics. The advantage of adding Fe₂O₃ to the material (K_{0.44}Na_{0.52}Li_{0.04}) (Nb_{0.84}Ta_{0.10}Sb_{0.06})O₃ (KNLNTS) s demonstrated in the paper [26]. additionally discussed in the article [27] concerning the incorporation of Fe_2O_3 into the KTaO₃/BaTiO₃ composite. However, the effect of adding Fe₂O₃ to BNT-ST has not been disclosed in these papers, possibly due to funding constraints and the fact that it is not the focus of their research. Based on the provided description, it seems that Fe₂O₃ would be a good doping material to enhance the electrical properties of BNT-ST.

Fe₂O₃ material can be synthesized from iron sand. Magnetic minerals contained in iron sand deposits are magnetite (Fe₃O₄), hematite (α -Fe₂O₃), and maghemite (γ -Fe₂O₃). Each mineral from iron sand has different properties. Maghemite has a cubic close-packed structure with good chemical equilibrium, while hematite has a rhombohedral structure and is the most stable type but is antiferromagnetic below the Neel temperature (<955 K). One of the iron sand potentials that can be optimized is located on Widara-

payung Beach, Binangun sub-district, Cilacap district. This sand contains magnetic minerals in the form of magnetite.

Iron sand is one source of natural magnetite or lodestone, which is a ferrous ferrite (Fe[Fe₂O₄], or Fe₃O₄) that occurs naturally [28]. The sintering process at high temperatures in making ceramic magnets results in a phase change from magnetite (Fe₃O₄) to maghemite (γ -Fe₂O₃). This change is followed by changes in the magnetic properties of the material [29]. Iron sand is needed in piezoelectric applications because it has the potential to increase the dielectric constant and stability of the piezoelectric material [26]. In general, it can be noted that BNT-ST is an alternative material for lead-free piezoelectrics. However, its piezoelectricity performance is still inferior compared to PZT which is currently widely used. On the other hand, Fe₂O₃ is known to increase the piezoelectric properties of a material. Thus, it is worth examining the characteristics of BNT-ST doped with Fe₂O₃, to obtain an environmentally friendly piezoelectric material that has excellent piezoelectricity. Furthermore, it is necessary to look for a BNT-ST-Fe₂O₃ composition that has the desired piezoelectricity.

 Fe_2O_3 addition to the BNT-ST piezoelectric material is a fascinating experiment that merits more investigation. This is consistent with the effectiveness of doing so as shown by studies [26] and [27]. Furthermore, it is necessary to identify the ideal composition of the Fe_2O_3 addition to produce the BNT-ST-Fe₂O₃ piezoelectric material, which has excellent electrical properties.

As per the above description, the material modification stage can be completed by altering the manufacturing process [18, 19] or the component material [20–22]. The material in this study was modified to make the necessary corrections. The material that was used in the present investigation to alter the BNT-ST piezoelectric material is called Fe₂O₃. Fe₂O₃, the donor ion doping material, has been shown to increase both the piezoelectric constant value [26] and the magnetoelectric response capability [27]. It was selected as a doping material in the BNT-ST piezoelectric material for these reasons. The next step is to formulate the problems that this study will attempt to solve.

Based on the background that has been described, the problem can be formulated as follows;

– what the characteristics of the Fe_2O_3 doped $Bi_{0.5}Na_{0.5}$ TiO₃-SrTiO₃ piezoelectric material are;

– what is the optimum composition of Fe_2O_3 doping for the $Bi_{0.5}Na_{0.5}TiO_3$ -SrTiO₃ piezoelectric material is.

3. The aim and objectives of the study

The aim of the study is to determine the optimum composition of the BNT-ST piezoelectric material doped with Fe_2O_3 to obtain a piezoelectric material that has a high dielectric constant, stable material, and low impedance.

To achieve this aim, the following objectives are accomplished:

- to synthesize $\rm Fe_2O_3$ doped $\rm Bi_{0.5}Na_{0.5}TiO_3$ -SrTiO_3 using the solid-state reaction method;

– to characterize the $\rm Fe_2O_3$ doped $\rm Bi_{0.5}Na_{0.5}TiO_3\text{-}Sr\text{-}TiO_3$ sample.

4. Materials and methods

The object of this research is the piezoelectric material of $Bi_{0.5}Na_{0.5}TiO_3$ -SrTiO₃ doped with Fe₂O₃.

The object-related hypothesis, which is based on the completed literature review, is that the BNT-ST piezoelectric material's electrical characteristics will be enhanced by the addition of Fe₂O₃. Furthermore, several assumptions are made concerning this study, specifically:

1. The vacuum permittivity value of 8.85×10^{-12} F/m is regarded as constant.

2. Fe₂O₃ synthesized independently from iron sand of the Binangun area, Cilacap is assumed to be equivalent to commercial Fe₂O₃. Fe₂O₃ is synthesized using the solid-state reaction method with a calcination temperature of 850 °C and a calcination time of 3 hours.

3. Because $Bi_{0.5}Na_{0.5}TiO_3$ -SrTiO₃ and Fe₂O₃ materials have been heated to 120 °C, it is expected that they are devoid of water.

Ensuing the following is the simplification used in the research project:

1. Piezoelectricity testing of Fe_2O_3 doped $Bi_{0.5}Na_{0.5}TiO_3$ -SrTiO₃ material was carried out at a frequency of 1000 Hz.

2. This study did not examine the output voltage.

3. Air pressure's impact was disregarded.

4. Piezoelectricity was tested by applying an electric field and without applying pressure.

The materials used in this study were $Bi(NO_3)_3$ 71 % Merck KGaA 1304-85-4 (source of Bi); Na₂CO₃ 99% Aldrich 497-19-8 (source of Na); TiO₂ 97 % Aldrich 13463-67-7 (source TiO₃); SrCO₃ 98 % Aldrich 1633-05-2 (source Sr); natural iron sand of Binangun area, Cilacap district, Central Java, Indonesia (source of Fe₂O₃); Aqua Demineralization; Merck absolute ethanol 64-17-5; and Merck KGaA 99 % polyvinyl alcohol. The appliances used are the PPF-UG FiLa shaker mill, PPF-1300 muffle furnace, Krisbow hydraulic pump, and PA 213 OHAUS digital balance. As well, X-Ray Diffraction (XRD) Philips PW 171016, Scanning Electron Microscope-Energy Dispersive Spectroscopy (SEM-EDS) JEOL type JSM-6510 LA Equipment, Inductor-Capacitor-Resistor meter (LCR meter) digital type 4271B. Synthesis of Bi_{0.5}Na_{0.5}TiO₃-SrTiO₃-Fe₂O₃ started by weighing Bi(NO₃)₃, Na₂CO₃, TiO₂, SrCO₃, and Fe₂O₃ based on the composition of the compound $(0.75-x)Bi_{0.5}Na_{0.5}TiO_3+(0.25-x)SrTiO_3+xFe_2O_3$ where x is 0; 0.025; 0.05; 0.075; and 0.1 in moles and are denoted by FeBS0, FeBS1, FeBS2, FeBS3, and FeBS4. Next, the samples were milled for 3 hours using a PPF-UG shaker mill with the addition of absolute ethanol and dried at a temperature of 120 °C. The dried sample was calcined in a furnace at a temperature of 850 °C to remove impurities in the sample. Then the sample was mashed again for 3 hours using a PPF UG shaker mill to reduce the surface area of the material so that the resulting pressure would be more optimal. The process of producing pellets is as follows. Five drops of 5 % polyvinyl alcohol were added to the sample to form a semi-dry powder (agglomerated powder). Furthermore, the material is ground for 5 minutes so that it is no longer agglomerated and is ready to be formed into pellets by using a Krisbow hydraulic pump. The sample was then pre-sintered for 1 hour at a temperature of 800 °C and sintered at 1100 °C for 2 hours. The material Bi_{0.5}Na_{0.5}TiO₃-SrTiO₃-Fe₂O₃ in pellets was tested for characterization and electrical properties, in the form of dielectric constant and impedance using an LCR meter with an increment of 100 °C. The samples were also tested using XRD (X-ray diffraction) and SEM (Scanning Electron Microscopy) apparatus for composition material and morphology analysis. The dielectric constant was determined using an LCR meter through a furnace. The LCR meter will evaluate

the capacitance of the prepared sample as the temperature of the furnace was raised, then the dielectric constant was calculated through the capacitance using the following equation:

$$\varepsilon_r = \frac{Cd}{\varepsilon_0 A},\tag{1}$$

where ε_r is the relative dielectric constant, *C* is capacitance (F), *d* is the distance between the plates (m), ε_0 is the air permittivity (8.85×10⁻¹² F/m), and *A* is the cross-sectional area of the plates.

5. Results of the synthesis and characterization of Fe₂O₃ doped Bi_{0.5}Na_{0.5}TiO₃-SrTiO₃

5. 1. Synthes of Fe₂O₃ doped Bi_{0.5}Na_{0.5}TiO₃-SrTiO₃ using the solid-state reaction method

The piezoelectric material Bi_{0.5}Na_{0.5}TiO₃-Sr-TiO₃ doped with Fe₂O₃ has been successfully synthesized using the solid-state reaction method. Fe₂O₃ was synthesized through the milling and calcination of iron sand from the Binangun area of Cilacap (Fe₃O₄), resulting in a material with a dark red color. This color signifies that the Fe₂O₃ formed is of the maghemite type (γ -Fe₂O₃). The synthesized Fe₂O₃ material was then used as a doping agent. The synthesis of Bi_{0.5}Na_{0.5}TiO₃-SrTiO₃ doped Fe₂O₃ using

the solid-state reaction method was performed by mixing Bi(NO₃)₃, Na₂CO₃, TiO₂, SrCO₃, and Fe₂O₃. Absolute ethanol was added to this mixture to ensure homogeneity during the milling process. The mixing process is followed by drying to remove water content and calcination to form the desired compound. This process yields a powder with a yellow color, which becomes progressively more brownish as the Fe₂O₃ concentration increases. This suggests the influence of Fe³⁺ ions, as evidenced by the characteristic color of Fe³⁺ salts within the material [25].

The powder is then ground using a mortar, a process that maximizes the contact surface area between particles. At this stage, 5 % PVA is added to the material, which serves to enhance its mechanical properties, providing strong resistance under high pressure and temperature conditions. The material is subsequently compacted under an external magnetic field, which aligns the spin direction within the crystals. The powder is compressed to form flat cylindrical pellets, which are then solidified and subjected to sintering for further heating, resulting in a coin pellet with a diameter of 1 cm and a thickness of 0.4 cm. The pure BNT-ST sample without doping is white, furthermore, the addition of Fe₂O₃ causes the color of the sample to change to yellow, and the more Fe₂O₃ added causes the color of the sample to change to brownish.

5. 2. Characterization of the Fe_2O_3 doped $Bi_{0.5}Na_{0.5}TiO_3$ -SrTiO₃ samples

The results of the LCR meter test are as follows. The dielectric constant is a factor that makes a specific substance reduce the electrostatic force between two separate charged objects under vacuum conditions. The magnitude of the value of the dielectric constant illustrates that the material can store electric charge along with one of the functions of the capacitor as a charge store. Testing the dielectric constant is carried out at various temperatures starting with room

temperature up to a temperature of 500 °C in a frequency of 1000 Hz using an LCR meter. The graph of the constant dielectric test against temperature can be seen in Fig. 1.



Fig. 1. Dielectric constant and Curie temperature of Fe_2O_3 doped $Bi_{0.5}Na_{0.5}TiO_3\text{-}SrTiO_3$

The sample that produces the optimum polarization ability is the sample of FeBS1 when compared to other concentration variations. Therefore, FeBS1 can have high electrical stability with an optimal temperature of 400 °C. The FeBS1 sample shows the high piezoelectricity which is characterized by the high dielectric constant. The sample that has the lowest piezoelectricity is the FeBS2 sample with a dielectric constant of 5,975 at a temperature of 30 °C. However, the FeBS2 sample still has an increase in the dielectric constant at temperatures of 100 °C and 200 °C, although the constant dielectric increase is not higher than the optimum condition of the FeBS2 sample. The effect of material impedance on temperature with a frequency of 1000 Hz is shown in each sample variation in Fig. 2.

All samples showed the highest impedance values in the range of 30 °C to 200 °C. Furthermore; the material impedance decreases at a temperature of 300 °C. It again increases at a temperature of 400 °C. In contrast to the dielectric constant, the relative dielectric constant has a low value at a temperature of 30 °C to 200 °C. However, at a temperature of 300 °C, the dielectric constant value will increase again, except for the FeBS2 sample. The summary of the results for the dielectric constant, Curie temperature, and impedance is presented in Table 1.

Next, the results of the XRD test are as follows (Fig. 3, 4).

Table 1

Summary of dielectric constant, temperature Curie, and impedance result of Fe_2O_3 doped $Bi_{0.5}Na_{0.5}TiO_3$ -SrTiO₃

Sample	Dielectric constant	Temperature Curie (°C)	Impedance (Ω)
FeBS0	7,214	300	147,370
FeBS1	12,037	400	139,280
FeBS2	7,185	30	146,500
FeBS3	5,975	200	145,320
FeBS4	12,029	300	142,770

The determination of the peaks in the diffraction pattern from the XRD results refers to the ICDD (International Center of Diffraction Data) database. $Bi_{0.5}Na_{0.5}TiO_3$ and SrTiO₃ phases were obtained from the XRD test results, which formed rhombohedral and cubic crystal structures in FeBS0 (codes 00-036-0340 and 00-040-1500). Furthermore, SrTiO₃, FeBi₅Ti₃O₁₅, and Na₂Ti₃O₇ phases appeared in FeBS1, FeBS2, FeBS3, and FeBS4 samples. The SrTiO₃, FeBi₅Ti₃O₁₅, and Na₂Ti₃O₇ phases are related to the cubic, orthorhombic, and monoclinic crystal structures (codes 00-040-1500, 00-038-1257, and 01-072-0148). The crystal size of each phase is calculated using the Debye-Scherner equation. Furthermore, the sample's microstructure was characterized using a Scanning Electron Microscope (SEM). Material characterization using SEM is used to see the structure and particle size. Fig. 5 is the SEM result of $Bi_{0.5}Na_{0.5}TiO_3$ -SrTiO₃ and Fe₂O₃ doped $Bi_{0.5}Na_{0.5}TiO_3$ -SrTiO₃ samples with a magnification of 3000 times.

Some parts of the Fe_2O_3 doped $Bi_{0.5}Na_{0.5}TiO_3$ -SrTiO₃sample still contain a polymer structure which is thought to be polyvinyl alcohol which has not yet been combined with other particles. The Samples have different particle sizes which is obtained by comparing the particle length with the length of the reference line located at the bottom right of Fig. 5. The particle size of each sample is shown in Table 1 below.





53







Fig. 5. Scanning Electron Microscopy images of samples: a - FeBS0; b - FeBS1; c - FeBS2; d - FeBS3; e - FeBS4

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	Table 2
Particle size of Fe_2O_3 doped	Bi _{0.5} Na _{0.5} TiO ₃ -SrTiO ₃

Sample	Particle size (µm)
FeBS0	1.17-2.64
FeBS1	0.88-2.64
FeBS2	1.17-3.23
FeBS3	1.47-5.88
FeBS4	2.58-8.23

The particle size that forms a range for each sample concentration variation may be caused by inhomogeneous collisions during the shaking process in the PPF-UG Shaker Mill. The sample that has the smallest particle size is FeBS1, while the sample that has the largest particle size is FeBS4.

6. Discussion of the effect of the Fe₂O₃ concentration doping on the characteristics of the BNT-ST piezoelectric material

In the LCR meter test results as shown in Table 1, the FeBS1 sample shows high piezoelectricity, characterized by the high dielectric constant. The high dielectric constant in the FeBS1 sample may be due to a higher number of sodium ions, which is characterized by a higher number of Na2Ti3O7 phases and a higher intensity. Sodium-ion is an ion with an electrical storage ability similar to lithium [30, 31]. The FeBS2 sample still has an increase in the dielectric constant at temperatures of 100 °C and 200 °C, although the constant dielectric increase is not higher than the optimum condition of the FeBS2 sample. The small value of piezoelectricity, in the form of dielectric constant and Curie temperature, could be due to the crystal size of FeBS2, which is the sample with the smallest crystal size among other sample variations. Apart from that, the small value of the dielectric constant and Curie temperature is also possible because the large value of dielectric loss due to the addition of iron ions does not have a more significant effect on FeBS2 compared to other concentration variations. In addition to the dielectric constant, the LCR meter can also be used for impedance analysis of materials. The FeBS2 sample has a material impedance that tends to be stable, resulting in a small dielectric constant. An impedance is taken to represent the dissipative component of the dielectric response. Therefore, the higher the impedance of the material, the smaller the value of the dielectric constant.

From the XRD examination data, the addition of iron is thought to have restructured $Bi_{0.5}Na_{0.5}TiO_3$ into $FeBi_5Ti_3O_{15}$ and $Na_2Ti_3O_7$. This phenomenon is caused by the small binding force of Na atoms, as evidenced by the electronegativity value of Na atoms, which is 0.93. The electronegativity value of Na atoms is smaller than that of Bi, Ti, and Fe atoms, namely 2.02, 1.54, and 1.83, which causes the Bi atom to bond first with the Fe and Ti atoms so that there is no leftover to share electrons with the Na atom.

The addition of iron ions from Fe₂O₃ has succeeded in doping Bi_{0.5}Na_{0.5}TiO₃, as evidenced by the addition of Fe in FeBi₅Ti₃O₁₅. The ionic radius of iron (0.67 Å), which is smaller than titanium (0.88 Å), allows the occurrence of interstitials between the Bi_{0.5}Na_{0.5}TiO₃ lattice and breaks down Bi_{0.5}Na_{0.5}TiO₃ compounds. It is suspected that the doped iron species is Fe³⁺. Meanwhile, the SrTiO₃ phase increased its peak intensity and did not undergo a phase change. The sintering treatment will make the ions in SrTiO₃ diffuse into the crystal structure without changing the oxidation number [32]. The more Fe₂O₃ doping concentration resulted in the decreasing intensity of the Na₂Ti₃O₇ phase and the increasing intensity of the SrTiO₃ and FeBi₅Ti₃O₁₅ phases. In the test results, the highest phase intensity was obtained in FeBS4.

FeBS0, FeBS1, FeBS2, FeBS3, and FeBS4 samples ideally produce larger crystal sizes as the doping composition increases. This is because the high-intensity value influences the magnitude of the FWHM value in each sample. The higher the intensity value, the smaller the FWHM value obtained because the width of the peak is getting smaller. Scherrer stated that the FWHM value is inversely proportional to the size of the crystal so a large crystal size is obtained for a small FWHM value. However, the crystal size in the sample composition of FeBS2 and FeBS3 tends to decrease more than the crystal size of FeBS1. FeBS2 and FeBS3 materials have an anomaly. This anomaly can be due to the bond in the sample is not homogeneous. In addition, the presence of the Na₂Ti₃O₇ phase, whose intensity decreases with some of the decreasing FeBi₅Ti₃O₁₅ phases, also can produce a small crystal size. However, there is also a SrTiO₃ phase and some other FeBi₅Ti₃O₁₅ phases whose intensity increases with doping.

All samples appear to have cavities or pores still, and some particles are fused. This structure indicates that the diffusion process in the sintering is not perfect, so the mechanical strength tends to be brittle. Some of the sides of the Fe₂O₃ doped Bi_{0.5}Na_{0.5}TiO₃-SrTiO₃ sample still have a polymer structure which is suspected to be polyvinyl alcohol that has not been fused with other particles. The size of the particles as shown in Table 1, that make up the range for each variation of the sample concentration indicates an inhomogeneous collision during the shaking process in the PPF-UG Shaker Mill.

Based on characterization data, both electrical properties data in the form of dielectric constant values and impedance values accompanied by curie temperature data; crystal phase and crystal size data as well as from the discussion presented, it appears that piezoelectric material doped with Fe_2O_3 in the amount of 0.025 mol % is the best composition for an alternative replacement for PZT.

The primary limitation of this study lies in the challenge of obtaining stable values with smaller temperature intervals when measuring the dielectric constant and Curie temperature using an LCR meter on materials within a furnace. Moreover, the inclusion of polyvinyl alcohol (PVA) during the grinding process appears to increase the pore area, consequently reducing the compactness of the ceramic samples. This observation is corroborated by the findings of [33], who demonstrated that the same method, with the addition of PVA as a binder incorporated differently, resulted in a smaller pore area and enhanced compactness. On the other hand, the disadvantage of this study is the increased porosity observed with higher concentrations of Fe₂O₃. A large area of porosity can limit the stability and strength of ceramic materials at high temperatures. When a material contains pores or voids, the overall volume of solid material contributing to the dielectric response is effectively reduced. The presence of air within the material creates regions with lower dielectric values. Porosity can be reduced by employing a fabrication method that includes higher sintering temperatures, increased compaction, and the incorporation of a lower concentration of PVA into the material. Furthermore, the results of this study indicate that, overall, the Bi_{0.5}Na_{0.5}TiO₃-SrTiO₃-Fe₂O₃ exhibits excellent dielectric properties. The scope of this research is the relative dielectric constant with distance between two plates, air permittivity, and a fixed cross-sectional area which is maintained constant, was evaluated as the temperature increases at intervals of 100 °C. Whilst, the scope of application of the results of this study is the use of Fe₂O₃ doped BNT-ST material as a piezoelectric material to replace lead-based piezoelectric so that it is secure for the environment. The limitations of the conditions for the application of the results of this study are the permittivity value used when calculating the dielectric constant is the permittivity value of air, not the permittivity value of other materials. The use of Fe₂O₃ doped BNT-ST piezoelectric material resulting from this study which has an increased dielectric constant value is expected to provide an increase in electrical gain when applied. Therefore, the study could be further developed using different fabrication methods and evaluated to determine the ferroelectric properties of the material.

7. Conclusions

1. Fe₂O₃ doped Bi_{0.5}Na_{0.5}TiO₃-SrTiO₃ had been successfully synthesized, resulting in a new compound in the form of FeBi₅Ti₃O₁₅-Na₂Ti₃O₇-SrTiO₃. The new compound has a crystal structure in cubic, orthorhombic, and monoclinic. The size of the crystallite tends to fluctuate with increasing concentration of Fe₂O₃, while the particle size starting from 0.88 µm tends to increase to 8.23 µm. The crystallite and particle size exhibited typical features of Fe³⁺ addition.

2. The more Fe₂O₃ doping concentration resulted in the decreasing intensity of the Na₂Ti₃O₇ phase and the increasing intensity of the SrTiO₃ and FeBi₅Ti₃O₁₅ phases. The optimum composition of Fe₂O₃ doped Bi_{0.5}Na_{0.5}TiO₃-SrTiO₃ is with Fe₂O₃ doping of 0.025 mol % because it has the highest dielectric constant (ε_r) and Curie temperature (*T_c*), that is 12.037 at *T_c*=400 °C and the most negligible ma-

terial impedance (Z) is $135 \text{ k}\Omega$. The obtained Curie temperature is significantly higher than that of the pure BNT. The results suggest that the synthesized material has the potential to exhibit superior piezoelectric properties compared to BNT.

Conflict of interest

The authors declare that they have no conflict of interest concerning this research, whether financial, personal, authorship or otherwise, that could affect the research and its results presented in this paper.

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Data availability

All data are available in the main text of the manuscript.

Use of artificial intelligence

The authors confirm that they did not use artificial intelligence technologies when creating the current work.

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